

Home Search Collections Journals About Contact us My IOPscience

Tunable dichroism and optical absorption of graphene by strain engineering

This content has been downloaded from IOPscience. Please scroll down to see the full text. 2015 2D Mater. 2 025001 (http://iopscience.iop.org/2053-1583/2/2/025001)

View the table of contents for this issue, or go to the journal homepage for more

Download details:

IP Address: 132.248.209.234 This content was downloaded on 29/07/2015 at 17:23

Please note that terms and conditions apply.

2D Materials

CrossMark

RECEIVED 11 November 2014

REVISED 30 January 2015

ACCEPTED FOR PUBLICATION 26 February 2015

PUBLISHED 2 April 2015

Tunable dichroism and optical absorption of graphene by strain engineering

M Oliva-Leyva^{1,2} and Gerardo G Naumis¹

¹ Depto. de Física-Química, Instituto de Física, Universidad Nacional Autónoma de México (UNAM), Apdo. Postal 20-364, 01000, México D.F., Mexico

² Author to whom any correspondence should be addressed.

E-mail: moliva@fisica.unam.mx and naumis@fisica.unam.mx

Keywords: graphene, strain engineering, optical absorption, dichroism

Abstract

PAPER

We theoretically study the transmittance for normal incidence of linearly polarized light between two media separated by a strained graphene monolayer. We analytically characterize the degree of dichroism and the transparency of graphene as a function of an arbitrary uniform strain and the incident polarization. We discuss how measurements of dichroism and transparency for two different polarization directions can be used to determine the magnitude and direction of strain. Our findings result in very useful tools to tune the graphene absorption by mechanical strain, as well as to design nano-devices to determine crack propagation in materials.

1. Introduction

Strain engineering has been widely used as an effective tool to improve the technological functionality of graphene. Its electrical, chemical and optical properties are highly sensitive to mechanical deformation because of an unusual interval of elastic response. Graphene can withstand a reversible stretching up to 20% [1].

Due to its relevance for graphene-based electronics, the band-gap opening in graphene is among the most investigated implications induced by deformations or electromagnetic fields [2-5]. Even though a uniaxial strain can be used to achieve the gap opening [2, 3], theoretical studies have predicted that a combination of strains results in more accessible set-ups [4]. Moreover, nonuniform deformations induce striking Landau levels due to effective pseudo-magnetic fields [6–9]. Perfect graphene has a low piezoresistive sensitivity, however, the graphene-based strain sensors are a promising field in nanotechnology [10-12]. In nanoand micro-electromechanical systems, the Casimir interaction is a unwanted problem. In recent study, it was theoretically investigated how the strain modifies the force Casimir in graphene-based systems [13]. On the other hand, a simple approach has been reported to control the chemical reactivity of graphene [14]. The application of strain, via stretching of the

supporting flexible substrate, produces impressive increases in the rate of reactivity [14].

Otherwise, the concept of strain engineering has been experimentally extended to the optical domain in recent works [15, 16]. Unstrained graphene (undoped) has a transparency of around 97.7% over a broad band of frequencies [17]. The origin of this remarkable feature, defined by fundamental constants, is ultimately a consequence of graphene's unique electronic structure. Needless to say, another strain effect is the anisotropy in the electronic dynamics [18], which is traduced in an anisotropic optical conductivity [19, 20]. Such effect for strained graphene yields a modulation of the transmittance as a function of the polarization direction. From a theoretical viewpoint, this modulation of the transmittance has been only quantified in the case of a uniaxial strain [21–24]. However, nowadays there are novel methods for applying biaxial strain in a controlled manner, even without the need for bending the substrate [25]. So, a more general theoretical characterization of the strained-graphene transparency is needed.

In this paper, we quantify the modulation of the transmittance for graphene under an arbitrary uniform strain (e.g., uniaxial, biaxial, and so forth), as a function of the polarization direction. Also, we characterize the degree of polarization rotation as a function of strain and polarization direction. These results



are useful to tune in effective manner the optical absorption of graphene, and hence, can be potentially utilized towards novel optical detectors, sensors, and photovoltaics.

2. Electromagnetic scattering problem

We concentrate on the transmittance for normal incidence of linearly polarized light between two media separated by a graphene sheet which is uniformly strained, as shown in figure 1(a). We assume that the media are characterized by the electrical permittivities $\epsilon_{1,2}$ and the magnetic permeabilities $\mu_{1,2}$. From figure 1(b), we observe that the electric (*E*) and magnetic (*H*) fields belong to the graphene plane while the incident and transmitted polarizations $\theta_{i,t}$ are measured with respect to the laboratory axes *xy*.

For this scattering problem, one can set down immediately the boundary conditions for the electromagnetic fields, which can be written as [26]

$$E_t - E_i - E_r = 0, (1)$$

$$n \times (H_t - H_i - H_r) = J, \qquad (2)$$

where J is the surface current density and n = (0, 0, 1). The electrical and magnetic field on each media are related by

$$H = \sqrt{\frac{\epsilon}{\mu}} \frac{k \times E}{k},\tag{3}$$

whereas Ohm's law reads

$$J = \bar{\sigma}(\omega) \cdot E_t, \tag{4}$$

with $\bar{\sigma}(\omega)$ being the frequency-dependent conductivity tensor of graphene. Under uniform strain, $\bar{\sigma}(\omega)$ is given by [19, 20]

$$\bar{\boldsymbol{\sigma}}(\omega) = \sigma_0(\omega) \Big(\bar{\boldsymbol{I}} - 2\tilde{\beta}\bar{\boldsymbol{\epsilon}} + \tilde{\beta}\mathrm{Tr}\left(\bar{\boldsymbol{\epsilon}}\right)\bar{\boldsymbol{I}} \Big), \qquad (5)$$

where \bar{I} is the 2 × 2 identity matrix, $\sigma_0(\omega)$ is the conductivity of unstrained graphene, \bar{e} is the strain tensor and $\tilde{\beta} \simeq 2.37$ is related to Grüneisen parameter. It is easy to see that an isotropic strain, $\bar{e} = e\bar{I}$, does not affect the conductivity whereas an anisotropic strain yields an anisotropic conductivity.

Combining (1)-(4), we obtain

$$E_{i} = \frac{1}{2} \sqrt{\frac{\mu_{1}}{\epsilon_{1}}} \left(\left(\sqrt{\frac{\epsilon_{1}}{\mu_{1}}} + \sqrt{\frac{\epsilon_{2}}{\mu_{2}}} \right) \bar{I} + \bar{\sigma} \right) \cdot E_{t}.$$
 (6)

Equations (5) and (6) shows how the strain-induced asymmetry of the conductivity tensor results in certain degree of dichroism [21]. Note that only for an isotropic conductivity (isotropic strain) the vectors E_i and E_t are collinear and then the dichroism disappears.

Now from (6) it is straightforward to write the transmittance as

$$T\left(\theta_{i}\right) \approx T_{0}\left(1 - \frac{2\sqrt{\mu_{1}\mu_{2}}}{\sqrt{\epsilon_{1}\mu_{2}} + \sqrt{\epsilon_{2}\mu_{1}}}\boldsymbol{e}_{i} \cdot \Re \bar{\boldsymbol{\sigma}} \cdot \boldsymbol{e}_{i}\right), (7)$$

where $e_i = (\cos \theta_i, \sin \theta_i)$ and T_0 is the transmittance for normal incidence between two media in absence of the graphene interface. The term, $e_i \cdot \Re \bar{\sigma} \cdot e_i$ $= \Re [\bar{\sigma}_{xx} \cos^2 \theta_i + \bar{\sigma}_{yy} \sin^2 \theta_i + \bar{\sigma}_{xy} \sin 2\theta_i]$, shows how an anisotropic absorbance yields the periodic modulation of the transmittance as a function of the polarization direction θ_i .

In order to illustrate such effects, dichroism and modulation of $T(\theta_i)$, let us assume both media to be vacuum ($\epsilon_{1,2} = \epsilon_0$ and $\mu_{1,2} = \mu_0$) and that the graphene is at half filling, i.e., the chemical potential equals to zero. In this case, for infrared and visible frequencies, $\sigma_0(\omega)$ is frequency-independent and is given by the universal value $e^2/(4\hbar)$. As a consequence, from (5)–(7) we obtain that the polarization angles $\theta_{i,t}$ are related by





$$\theta_t - \theta_i \approx \alpha \tilde{\beta} \left(\frac{\bar{\epsilon}_{yy} - \bar{\epsilon}_{xx}}{2} \sin 2\theta_i + \bar{\epsilon}_{xy} \cos 2\theta_i \right) 180^\circ, \quad (8)$$

whereas the transmittance results in

$$T\left(\theta_{i}\right) \approx 1 - \pi \alpha \left(1 - \tilde{\beta} \left(\bar{\epsilon}_{xx} - \bar{\epsilon}_{yy}\right) \times \cos 2\theta_{i} - 2\tilde{\beta}\bar{\epsilon}_{xy} \sin 2\theta_{i}\right), \qquad (9)$$

where α is the fine-structure constant. Now it is easy to see the periodic modulations of the dichroism and transmittance, with a period of 180°, which is a simple consequence of the physical equivalence between the polarization angles θ_i and $\theta_i + 180°$ for normal incidence of linearly polarized light.

3. Discussion and conclusions

Let us explore first, the consequences on dichroism. In figure 2(a), we present the difference $\theta_t - \theta_i$ for two different deformations: a uniaxial deformation and a uniaxial-shear deformation. The resulting modulations are out of phase with each other because the principal strain directions of both deformations do not match. For the uniaxial case considered, the principal strain directions match the laboratory axes xy, and $\theta_t - \theta_i$ displays a sin $2\theta_i$ -like modulation (see blue line). On the other hand, for the uniaxial-shear case, its principal strain directions do not match the laboratory axes xy, and thus $\theta_t - \theta_i$ results in a $\sim \sin (2\theta_i - \phi)$ -like modulation (see red line), with $\tan \phi = 2\bar{\epsilon}_{xy}/(\bar{\epsilon}_{xx} - \bar{\epsilon}_{yy})$. This behavior shows that the principal strain directions can be determined by measuring the polarization angles θ_i for which the incident and transmitted polarizations coincide. It is important to note that the polarization angles can be routinely measured with a precision of 0.001°, so that the predicted modulation obtained from (8) can be experimentally monitored.

Finally, let us make some important remarks about our formula (9) concerning transmittance. First of all, one can see that (9) reduces to the universal value $1 - \pi \alpha \approx 97.7\%$, which is the transmittance of unstrained or isotropically strained graphene [17].

Likewise, (9) reproduces the previously reported modulation, $1 - \pi \alpha (1 - \tilde{\beta} (1 + \nu)\epsilon \cos 2\theta_i)$, for the case of a uniaxial strain along the *x*-axis, where ϵ is the strain magnitude and ν is the Poisson ratio ($\nu \approx 0.16$). In recent experiments, this periodic modulation of the transmittance for graphene has been confirmed [15]. In this experimental study, the strain magnitude ϵ was estimated from the modulation amplitude measurements by means of $\epsilon \approx \Delta T/(2\pi\tilde{\beta} (1 + \nu))$ and confirmed from the Raman spectroscopy measurements [15].

Thus, (9) contains all previously studied limiting cases. Moreover, it can be used in the more general scenario of biaxial strain. In fact, it provides a simple and reliable protocol to reconstruct the principal axes of the strain tensor. The protocol goes as follows

(1) measure the transmittance at $\theta_i = 0^\circ$,

(2) measure the transmittance at $\theta_i = 45^\circ$,

(3) from (9),

$$\tan \phi = \frac{1 - \pi \alpha - T(45^{\circ})}{1 - \pi \alpha - T(0^{\circ})},$$
 (10)

(4) then ϕ is just the angle between the laboratory axes *xy* and the principal strain directions.

In conclusion, we calculated the dichroism and the transparency for normal incidence of linearly polarized light between two media separated by a graphene monolayer under any arbitrary uniform strain. Our results contained some previously found particular cases. Then we proposed a new protocol based on two simple transmittance measurements to reconstruct the applied strain field, and in particular, the principal axes of strain. Such a protocol can be extremely useful to produce nano-sensors capable of detecting the local principal axis of strain, which are in fact determinant to determine crack propagation in different kinds of materials. Also, it can serve to measure pseudo-magnetic fields associated with graphene.

Acknowledgments

We thank the DGAPA-UNAM project IN-102513. MO-L acknowledges a scholarship from CONACyT (Mexico).

References

- Lee C, Wei X, Kysar J W and Hone J 2008 Measurement of the elastic properties and intrinsic strength of monolayer graphene Science 321 385–8
- [2] Pereira V M, Neto A H C and Peres N M R 2009 Tight-binding approach to uniaxial strain in graphene Phys. Rev. B 80 045401
- [3] Ni H Z, Yu T, Lu Y H, Wang Y Y, Feng Y P and Shen Z X 2008 Uniaxial strain on graphene: Raman spectroscopy study and band-gap opening ACS Nano 2 2301–5
- [4] Cocco G, Cadelano E and Colombo L 2010 Gap opening in graphene by shear strain *Phys. Rev.* B **81** 241412
- [5] López-Rodriguez F J and Naumis G G 2008 Analytic solution for electrons and holes in graphene under electromagnetic waves: gap appearance and nonlinear effects *Phys. Rev.* B 78 201406
- [6] Guinea F, Katsnelson M I and Geim A K 2010 Energy gaps and a zero-field quantum hall effect in graphene by strain engineering *Nat. Phys.* 6 30–33
- [7] Levy N, Burke S A, Meaker K L, Panlasigui M, Zettl A, Guinea F, Castro Neto A H and Crommie M F 2010 Straininduced pseudo-magnetic fields greater than 300 tesla in graphene nanobubbles *Science* 329 544–7
- [8] Sloan J V, Sanjuan A A P, Wang Z, Horvath C and Barraza-Lopez S 2013 Strain gauge fields for rippled graphene membranes under central mechanical load: an approach beyond first-order continuum elasticity *Phys. Rev.* B 87 155436
- [9] Qi Z, Kitt A L, Park H S, Pereira V M, Campbell D K and Neto A H C 2014 Pseudomagnetic fields in graphene nanobubbles of constrained geometry: a molecular dynamics study *Phys. Rev.* B 90 125419
- [10] Bae S-H, Lee Y, Sharma B K, Lee H-J, Kim J-H and Ahn J-H 2013 Graphene-based transparent strain sensor *Carbon* 51 236–42
- [11] Zhu S-E, Ghatkesar M K, Zhang C and Janssen G C A M 2013 Graphene based piezoresistive pressure sensor Appl. Phys. Lett. 102 161904

- [12] Jing Z, Guang-Yu Z and Dong-Xia S 2013 Review of graphenebased strain sensors *Chin. Phys.* B 22 057701
- [13] Phan A D and Phan T-L 2014 Casimir interactions in strained graphene systems Phys. Status Solidi-Rapid Res. Lett. 8 1003–6
- [14] Bissett M A, Konabe S, Okada S, Tsuji M and Ago H 2013 Enhanced chemical reactivity of graphene induced by mechanical strain ACS Nano 7 10335–43
- [15] Ni G-X, Yang H-Z, Ji W, Baeck S-J, Toh C-T, Ahn J-H, Pereira V M and Özyilmaz B 2014 Tuning optical conductivity of large-scale cvd graphene by strain engineering *Adv. Mater.* 26 1081–6
- [16] Dong B, Wang P, Liu Z-B, Chen X-D, Jiang W-S, Xin W, Xing F and Tian J-G 2014 Large tunable optical absorption of cvd graphene under total internal reflection by strain engineering *Nanotechnology* 25 455707
- [17] Nair R R, Blake P, Grigorenko A N, Novoselov K S, Booth T J, Stauber T, Peres N M R and Geim A K 2008 Fine structure constant defines visual transparency of graphene *Science* 320 1308
- [18] Oliva-Leyva M and Naumis G G 2013 Understanding electron behavior in strained graphene as a reciprocal space distortion *Phys. Rev.* B 88 085430
- [19] Oliva-Leyva M and Naumis G G 2014 Anisotropic ac conductivity of strained graphene J. Phys.: Condens. Matter 26 125302
- [20] Oliva-Leyva M and Naumis G G 2014 Corrigendum: Anisotropic ac conductivity of strained graphene J. Phys.: Condens. Matter 26 279501
- [21] Pereira V M, Ribeiro R M, Peres N M R and Neto A H C 2010 Optical properties of strained graphene *Europhys. Lett.* 92 67001
- [22] Pellegrino F M D, Angilella G G N and Pucci R 2009 Effect of uniaxial strain on the reflectivity of graphene *High Press. Res.* 29 569–72
- [23] Pellegrino F M D, Angilella G G N and Pucci R 2010 Strain effect on the optical conductivity of graphene Phys. Rev. B 81 035411
- [24] Pellegrino F M D, Angilella G G N and Pucci R 2011 Linear response correlation functions in strained graphene *Phys. Rev.* B 84 195407
- [25] Shioya H, Craciun M F, Russo S, Yamamoto M and Tarucha S 2014 Straining graphene using thin film shrinkage methods *Nano Lett.* 141158–63
- [26] Jackson J D 1999 Classical Electrodynamics 3rd edn (New York: Wiley) p 16